

High-temperature Electrical Behavior of a 2D Multilayered MoS₂ Transistor

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This paper reports the high-temperature-dependent electrical behavior of a 2D multilayer MoS₂ transistor. The existence of a big Schottky barrier at the MoS₂-Ti junction can reduce carrier transport and lead to a lower transistor conductance. At a high temperature (380 K), the field-effect mobility of the multilayer MoS₂ transistor increases to 16.9 cm²V⁻¹sec⁻¹, which is 2 times higher than the value at room temperature. These results demonstrate that at high temperature, carrier transport in a MoS₂ with a high Schottky barrier is mainly affected by thermionic emission over the energy barrier.

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I. INTRODUCTION

For future display technologies, there are strong demands to provide high-resolution and mechanically-flexible flat panels, and significant progress has recently been achieved in improving high-performance flexible and foldable organic light-emitting diode (OLED) displays [1,2]. A conventional display consists of two major systems: a backplane to drive the OLED display device and address active-matrix pixels and a display device to create display images and movies. Sustained efforts to realize flexible backplanes have been reported for flexible and conformable thin-film transistors (TFTs), a core device to drive/switch active matrix display pixels [3–5]. However, conventional thin-film materials such as amorphous Si (α -Si), low-temperature poly Si (LTPS), and oxides, limit the use of such TFTs in flexible backplane-circuitry due to their fragility and relatively low mobility.

Two-dimensional (2D) layered semiconducting chalcogenides (such as MoS₂, MoSe₂, WS₂, WSe₂, *etc.*) have attracted much attention due to their having an intrinsically high carrier mobility (> 100 cm²V⁻¹s⁻¹), mechanical flexibility for a 2D layered structure, and a finite bandgap (~ 1.35 eV) [6–8]. For a high-performance 2D layered semiconducting transistor, recent reports suggest that Ti was selected from among the low work function materials because its favorable interface geometry Ti allows bonding and the electronic density of states (DOS) at the Fermi level to be maximized through an increase in the overlap between the states at the interface [9]. However, such improvements for the transistors to date have fundamentally been hampered by the presence of a

Schottky barrier at the 2D layered MoS₂-Ti metal junction. A natural bulk MoS₂ crystal (SPI supplies, USA) is typically found in the ‘2H phase’, what has a semiconducting property, but often contains the ‘3R phase’, what shows a semi-metallic behavior [10]. This is because of the doping concentration and the chirality in the natural MoS₂ crystal, which can not to be controlled exactly. Previous reports on MoS₂ transistors show outstanding electrical properties, including high performance and a low subthreshold slope [4], but some of those devices fabricated till to date have contained a high Schottky barrier at the MoS₂-Ti metal junction after H₂ annealing, and the presence of the Schottky barriers severely blocks carrier transport in the channel and limits the transistor performance matrix.

In this research, we investigated the high-temperature electrical behavior of a MoS₂ transistor with a high Schottky barrier. From experimental temperature-dependent current-voltage (IV) measurements, the extracted field-effect mobility of the MoS₂ transistor was found to be proportional to the temperature. The carrier mobility in a typical MoS₂ crystal is limited by optical phonon scattering at high temperature, but the present immature MoS₂ transistors, as opposed to ohmically-contacted MoS₂ transistors, show an enhanced mobility at high temperature. High temperature leads to a larger thermionic emission that transports electrons over the energy barrier. Furthermore, the carrier transport mechanism and the calculation of Schottky barrier in the multilayer MoS₂ devices are discussed based on the observations.

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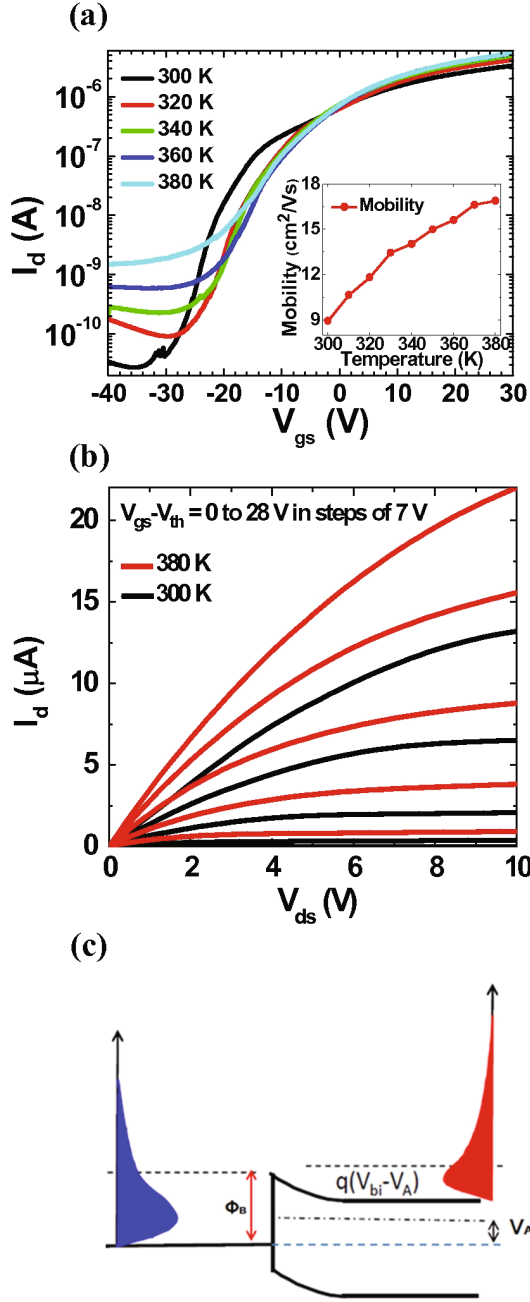


Fig. 3. (Color online) Electrical characteristics of a 2D multilayer MoS₂ transistor for various temperatures. (a) The I_d - V_{gs} characteristics of a representative device. The inset shows the temperature-dependent field-effect mobility at temperatures from 300 K to 380 K in steps of 10 K. (b) I_d - V_{gs} characteristics for $V_{gs}-V_{th}$ at room temperature (black line) and high temperature of 380 K (red line). (c) Energy band diagram and carrier distribution at the Schottky barrier between the MoS₂ semiconductor and the Ti/Au metal.

optical-phonon and acoustic-phonon scattering above room temperature. Thus, the extracted field-effect mobility of a good device should be inversely proportional to the temperature. However, the mobility behavior of the

device (Fig. 3(a)) is fully contradictory with that of the previous good device; the mobility of the present MoS₂ transistor ($\mu_m < 10 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$) is proportional to the temperature. At high temperature, the lattice vibration limits the carrier mobility at the ohmically-contacted 2D MoS₂ transistor, but the transport of electrons on this device is strongly limited by the thermionic emission over the potential barrier due to the large Schottky barrier (Φ_B) at MoS₂-Ti/Au junction. The current due to thermionic emission in a representative MoS₂ transistor, as shown in Fig. 3(c), is given by

$$J_T = A * T^2 * e^{-\frac{q\Phi_B}{kT}} [e^{\frac{qV_A}{kT}} - 1], \quad (1)$$

where Φ_B is the Schottky barrier height, V_A is an applied bias, and the parameter A is the effective Richardson constant. The above equation shows that the probability of carriers going over the potential barrier is enormously increased as the temperature is increased, thus, a larger total current (J_T) leads to a higher mobility at higher temperatures. The carrier transport in the present transistor with a high Schottky barrier is strongly limited by thermionic emission; thus, the mobility at high temperature (380 K) is increased to $16.9 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$, which is 2 times higher than that at room temperature.

IV. CONCLUSION

In conclusion, the high-temperature-dependent carrier transport of a MoS₂ transistor with a high contact resistance is investigated. By comparing the transistor mobility with increasing operation temperature, we found the μ_m to be limited by thermionic emission at high operation temperature due to the high Schottky barrier at the MoS₂-Ti/Au metal junction. In this regard, a high-performance 2D multilayer MoS₂ transistor having a zero or slightly negative Schottky barrier can be realized. Furthermore, the observing the high-temperature electrical behavior will be a good method for determining whether the device is an ohmically-contacted or a large-Schottky-barrier device.

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REFERENCES

- [1] S. Kim, H. J. Kwon, S. Lee, H. Shim, Y. Chun, W. Choi, J. Kwack, D. Han, M. Song, S. Kim, S. Mohammadi, I. Kee and S. Y. Lee, *Adv. Mater.* **23**, 3511 (2011)

- [2] H. Sasabe, J. I. Takamatsu, T. Motoyama, S. Watanabe, G. Wagenblast, N. Langer, O. Molt, E. Fuchs, C. Lennartz and J. Kido, *Adv. Mater.* **22**, 5003 (2010)
- [3] Q. Cao, H. S. Kim, N. Pimparkar, J. P. Kulkarni, C. Wang, M. Shim, K. Roy, M. A. Alam and J. A. Rogers, *Nature.* **454**, 495 (2008)
- [4] S. Kim, A. Konar, W. S. Hwang, J. H. Lee, J. Lee, J. Yang, C. Jung, H. Kim, J. B. Yoo, J. Y. Choi, Y. W. Jin, S. Y. Lee, D. Jena, W. Choi and K. Kim, *Nat. Commun.* **3**, 1011 (2012)
- [5] J. Lee, D. H. Kim, J. Y. Kim, B. Yoo, J. W. Chung, J. I. Park, B. L. Lee, J. Y. Jung, J. S. Park, B. Koo, S. Im, J. W. Kim, B. Song, M. H. Jung, J. E. Jang, Y. W. Jin and S. Y. Lee, *Adv. Mater.* **25**, 5886 (2013)
- [6] W. Choi, M. Y. Cho, A. Konar, J. H. Lee, G. B. Cha, S. C. Hong, S. Kim, J. Kim, D. Jena, J. Joo and S. Kim, *Adv. Mater.* **24**, 5832 (2012)
- [7] RadisavljevicB, RadenovicA, BrivioJ, GiacomettiV and KisA, *Nat Nano.* **6**, 147 (2011)
- [8] J. Pu, Y. Yomogida, K. K. Liu, L. J. Li, Y. Iwasa and T. Takenobu, *Nano Letters.* **12**, 4013 (2012)
- [9] K. F. Mak, C. Lee, J. Hone, J. Shan and T. F. Heinz, *Physical Review Letters.* **105**, 136805 (2010)
- [10] M. Chhowalla, H. S. Shin, G. Eda, L. J. Li, K. P. Loh and H. Zhang, *Nat Chem.* **5**, 263 (2013)