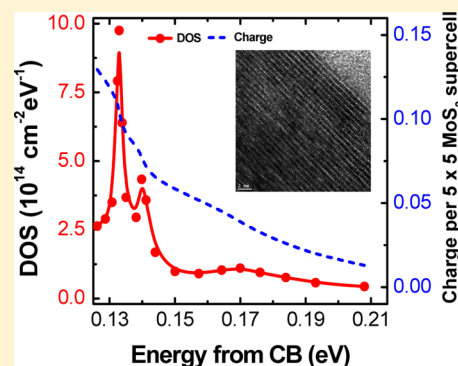


# Rendering High Charge Density of States in Ionic Liquid-Gated MoS<sub>2</sub> Transistors

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**ABSTRACT:** We investigated high charge density of states (DOS) in the bandgap of MoS<sub>2</sub> nanosheets with variable temperature measurements on ionic liquid-gated MoS<sub>2</sub> transistors. The thermally activated charge transport indicates that the electrical current in the two-dimensional MoS<sub>2</sub> nanosheets under high charge density state follows the Meyer-Neldel rule. The achieved high charge density allows the surface DOS estimation of MoS<sub>2</sub> nanosheets, which have distinct peaks at 0.135 and 0.145 eV below the conduction band, with the largest DOS values of  $9.75 \times 10^{14}$  and  $4.33 \times 10^{14} \text{ cm}^{-2} \text{ eV}^{-1}$ , respectively. This may represent the monolayer MoS<sub>2</sub> nanosheets that coexist with the MoS<sub>2</sub> multilayer in the channel area of the ionic liquid-gated MoS<sub>2</sub> transistors.



## 1. INTRODUCTION

Since the discovery of graphene in 2004, two-dimensional (2D) nanosheet materials have been widely explored as promising candidates for postsilicon semiconducting materials due to their potential for future electronic applications, such as high speed semiconductor chips on flexible substrates and transparent optoelectronics applications.<sup>1,2</sup> In particular, the benchmark player of 2D materials, graphene, shows excellent intrinsic optoelectronic properties and mechanical compatibility with plastic substrates, which are desirable for flexible electronics. Although 2D graphene has these exciting optical and mechanical properties, the zero-band gap in its electronic structure (a very low on/off current ratio  $< 100$ ) makes it unsuitable for a switching device in integrated circuitry.<sup>2–5</sup>

Alternately, the transition metal dichalcogenide molybdenum disulfide (MoS<sub>2</sub>), consisting of a Mo atom layer sandwiched between two sulfurs, has attracted great interest because it has electronic properties that are comparable to those of graphene but with sizable band-gaps due to its asymmetry at the atomic level.<sup>6,7</sup> Motivated by the distinct electronic properties of MoS<sub>2</sub> materials, we previously presented process-friendly multilayer MoS<sub>2</sub> thin-film transistors with high mobility of  $>100 \text{ cm}^2/(\text{V s})$  and robust current saturation over a large voltage window.<sup>7</sup> Very recently, Perera et al. also reported few-layer MoS<sub>2</sub>-based transistors showing a high electron mobility of  $60 \text{ cm}^2/(\text{V s})$ , a high on/off current ratio of  $10^7$  and near-ideal subthreshold swings of  $\sim 50 \text{ mV}$  per decade with the help of electrolyte gating.<sup>8</sup> However, despite these intensive studies on the device's performance, the fundamental behaviors underlying the charge transport phenomena on the surface of the MoS<sub>2</sub> nanosheet remain controversial. This is mainly attributed to the

lack of an experimentally proven study examining its electronic structures such as the density of states (DOS), which is essential to understand the intrinsic charge transporting characteristics.

In this article, we experimentally extracted the DOS in the MoS<sub>2</sub> nanosheet with electrostatic double layer (EDL)-gated transistors (EDLTs) from the experimental temperature-dependent current–voltage (IV) measurement, consisting of source and drain electrodes and an MoS<sub>2</sub> channel interfacing with a gate electrode in ionic contact using electrolytes. EDLTs using a highly capacitive EDL have significant advantages over conventional dielectric-based transistors in terms of the enhanced drain current, low operating voltages, new device configurations (e.g., coplanar gate structures), and the solution processability of solid electrolytes.<sup>9,10</sup> In particular, ionic liquids (ILs), which are composed of large organic cations and weakly coordinating anions, have received a great deal of attention as electrolytes due to their high ionic conductivity, wide electrochemical window, low vapor pressures, and chemical and physical stability.<sup>11</sup> Along with the nonvolatility and thermal stability, these characteristics allow ILs to reliably operate EDLTs at high temperatures ( $>100 \text{ }^\circ\text{C}$ ) through a thermally activated ion transport.<sup>10–13</sup>

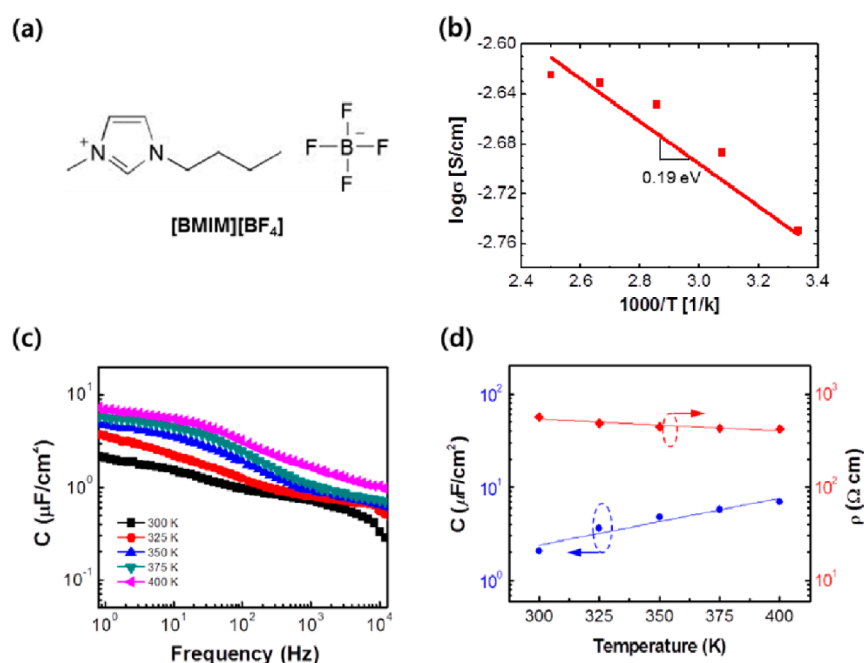
## 2. EXPERIMENTAL SECTION

**Device Fabrication.** The MoS<sub>2</sub> nanosheet-based EDLT devices were fabricated with source/drain electrodes, Ti(20

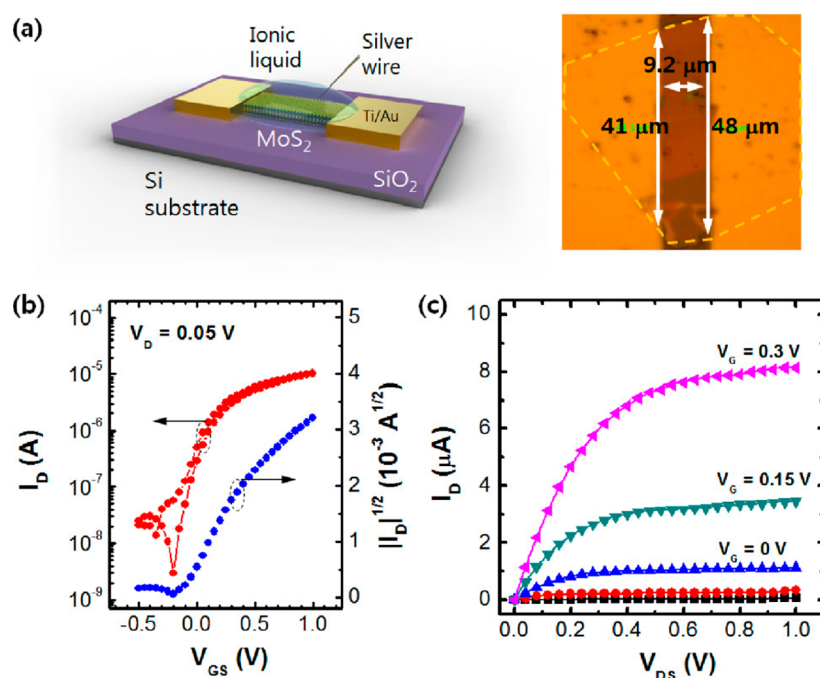
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**Figure 1.** (a) Molecular structures of 1-butyl-3-methyl-imidazolium tetrafluoroborate (BMimBF<sub>4</sub>) ionic liquids (ILs) used in this work. (b) Complex plane of the BMimBF<sub>4</sub> ILs measured from the impedance spectroscopy, (c) capacitance vs. frequency plot, and (d) summarized plot of the temperature-dependent capacitance and resistivity of the BMimBF<sub>4</sub> ILs.

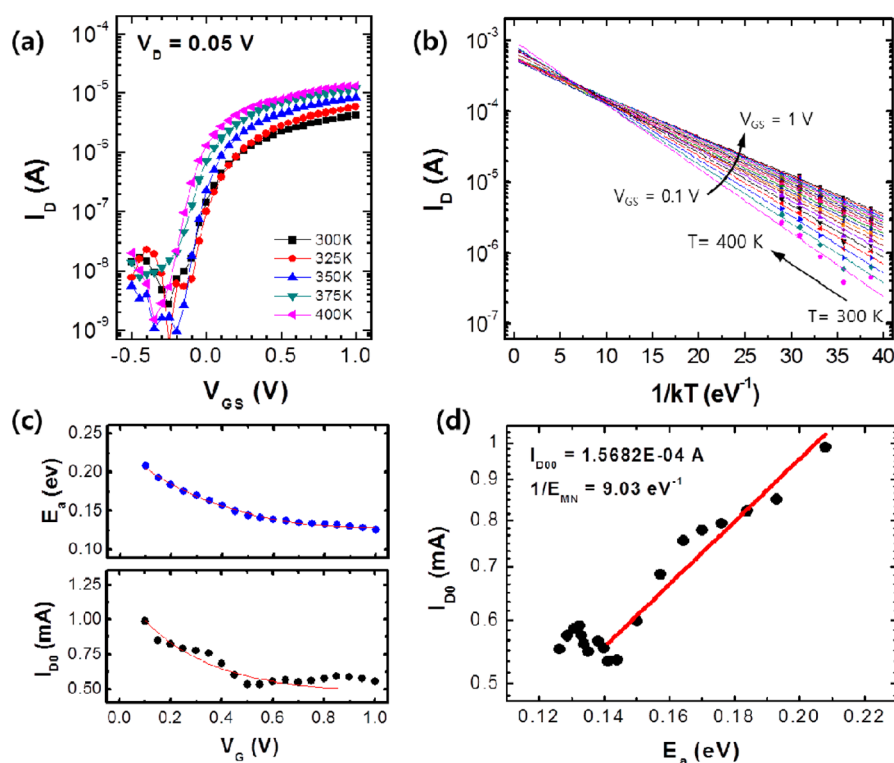


**Figure 2.** A MoS<sub>2</sub> nanosheet-based EDLT with IL gating on a SiO<sub>2</sub>/Si substrate and its electrical characteristics. (a) Schematic depiction and optical photographic image of the device. (b) The linear scale and log scale  $I_D$ - $V_{gs}$  characteristics and (c) output curves from the MoS<sub>2</sub> EDLT.

nm)/Au(300 nm), which were deposited by an electron beam evaporator. Thin flakes of MoS<sub>2</sub> were mechanically exfoliated from bulk MoS<sub>2</sub> crystals using Scotch tape and were then deposited on an SiO<sub>2</sub> (300 nm) dielectric and Si substrate as reported in our previous work.<sup>7</sup> Large-scaled stripes of Ti/Au electrodes (0.2 × 1.5 cm) are patterned by using standard photolithography, which makes it easy to align the PDMS structure on the channel for the IL droplets. 1-Butyl-3-methylimidazolium tetrafluoroborate (BMimBF<sub>4</sub>) was used as dielectric ILs. BMimBF<sub>4</sub> ILs with sizes of ~50 μL are dropped

onto the channel in the rectangular form of the PDMS to prevent the leakage of liquid electrolytes between the MoS<sub>2</sub> channel and the source/drain electrodes. For the electrolyte gating, silver wire was directly immersed in the IL gate medium and served as the top gate. The channel dimensions estimated by an optical microscope included a length of 41.1 μm and a width of 9.2 μm.

**Characterization.** The ionic conductivity and capacitance of ILs was measured by the AC impedance spectroscopy (a Solatron 1255 frequency response analyzer) over a frequency



**Figure 3.** (a) Transfer characteristics of the MoS<sub>2</sub>-based EDLT devices measured at various temperatures, ranging from 300 to 400 K. (b) Temperature dependence of the drain current,  $I_D$ . Scattered points represent the measured data, and the lines are fitted data to extract  $E_a$ . (c) Activation energy ( $E_a$ ) and prefactor ( $I_{D0}$ ) as a function of  $V_G$ . (d)  $I_{D0}$  vs  $E_a$  plot to estimate the Meyer-Neldel parameter ( $E_{MN}$ ).

range from  $10^{-1}$  Hz to  $10^6$  Hz. The homemade metal electrode–insulator–metal electrode (MIM) cells were configured for the measurement of the ionic conductivity and capacitance. BMimBF<sub>4</sub> ILs (used as insulating materials) were inserted into a 1 cm wide space of MIM cells between two Ti/Au electrodes with a contact area of 1 cm<sup>2</sup>. The cell was made up of PTFE to prevent melting or unexpected side reactions at elevated temperatures. The electrical EDLT characterization was performed with a semiconductor characterization system (Keithley 4200 SCS) connected to the probe station.

### 3. RESULTS AND DISCUSSION

Figure 1a shows the chemical structure of BMimBF<sub>4</sub>, which is the representative imidazolium family IL due to their relatively high ionic conductivity and thermal stability.<sup>11–13</sup> In fact, for EDLTs, the capacitive behavior and ionic transport of the IL electrolytes significantly influence the device features. Thus, we confirmed the temperature dependence of BMimBF<sub>4</sub> in the range of 300–400 K by impedance spectroscopy. As shown in Figure 1b, the ionic conductivities of ILs were enhanced at elevated temperatures due to the facilitated ion transport. The activation energy for ionic conduction, which can be derived from the Arrhenius equation, was determined to be 0.19 eV, indicating an ion hopping mechanism through acid–base pairs between protonated imidazolium rings and tetrafluoroborate anions.<sup>14</sup> The DC capacitances of Figure 1c in the BMimBF<sub>4</sub> ILs were calculated at different temperatures using the equation:  $C = -1/2\pi fZ''$ , where  $C$  is the DC capacitance,  $f$  is the frequency (in a range from  $10^{-1}$  to  $10^6$  Hz), and  $Z''$  is the imaginary impedance. The calculated capacitance value at 300 K was  $3.63 \mu\text{F}/\text{cm}^2$  (at a frequency of  $\sim 10^{-1}$  Hz), which is in good agreement with previous literature<sup>15</sup> and is about 3 orders

of magnitude greater than that of a several hundred nanometer thick SiO<sub>2</sub> dielectric. Importantly, the capacitance value was enhanced as a function of temperature due to the thermally activated ion transport. As a result, the DC capacitances at 400 K at the same frequency were improved up to  $12.4 \mu\text{F}/\text{cm}^2$ , which is three times greater than the value at 300 K. In a similar manner, we also estimate the resistivity values at different temperatures. Even above 80 °C, typical polymer electrolytes drastically change values due to the evaporation of the conduction media (i.e., water),<sup>16</sup> but the resistivity of the BMimBF<sub>4</sub> ILs is still consistent due to the humidity-independent transport mechanism as shown in Figure 1d (combined with capacitance). This finding should be addressed again because the thermally stable ion transports of the ILs would ensure that the EDLT system can be properly operated at the elevated temperatures.

Figure 2a illustrates a schematic device configuration of the MoS<sub>2</sub> nanosheet-based EDLT with IL. As described in the Experimental Section, Ti/Au electrodes and BMimBF<sub>4</sub> were used as source/drain electrodes and dielectric, respectively. Figure 2b displays the measured transfer characteristics (source-drain current,  $I_{DS}$ , as a function of the gate voltage,  $V_G$ ) of the MoS<sub>2</sub> EDLTs at a fixed source-drain voltage,  $V_{DS}$ , of 0.05 V. The source–drain current  $I_{DS}$  increased in proportion to the gate voltage ( $V_G$ ) in the positive direction up to 1 V. This behavior of MoS<sub>2</sub> EDLTs corresponds to the typical characteristics of *n*-type transistors, where the Fermi level is deep in the conduction band. Although the slight hysteresis of MoS<sub>2</sub> EDLTs was observed at this scanning speed (12.5 mV/s) of gate bias, the feature of the transfer curve was consistent in both the forward and reverse scans of  $V_G$ . The threshold voltage ( $V_T$ ) was  $-0.2$  V. This low-operation voltage has the advantage of exceptionally high capacitance values for both

EDL and the 2D-nanosheet channel. As further supported by a log-scale plot of  $I_{DS}$  near the onset of electron conduction, the subthreshold swing of 80 mV/decade for the electrons was attributed to the coupling effect of the ion gating and the 2D MoS<sub>2</sub> electronics, and the high quality of the MoS<sub>2</sub>/IL interface. Two electrostatic double layers at the metal/electrolyte and semiconductor/electrolyte are rapidly constructed due to the fast ion mobility and high capacitance of the ILs. Due to the capacitive behavior of the IL gating, a sufficient carrier concentration in the MoS<sub>2</sub> channel is readily accumulated. The on/off current ratio is  $3.36 \times 10^3$ , which is greater than that of the previously reported MoS<sub>2</sub> EDLTs gated by ILs. The electron mobility of the MoS<sub>2</sub>-based EDLTs in the linear regime was  $13.3 \text{ cm}^2/(\text{V s})$ , which was estimated with the following equation:<sup>14</sup>

$$I_D = \mu W/LC_{EDL}(V_G - V_T)V_D \quad (1)$$

where  $L$  is the channel length of  $9.2 \mu\text{m}$ ,  $W$  is the channel width of  $41 \mu\text{m}$ , and  $C_{EDL}$  is the capacitance of the dielectric. It should be noted that we used the capacitance value obtained from the impedance spectra at 0.1 Hz because it is assumed to be closer to static behavior than dynamic. The  $I_{DS}$ - $V_{DS}$  characteristics of the MoS<sub>2</sub> EDLTs with a gate voltage,  $V_G$ , in the range from  $-0.3$  to  $0.3$  V are also presented in Figure 2c. The  $I_{DS}$  was linearly proportional to  $V_{DS}$  below a  $V_G$  of 0.5 V, indicating that the source and drain electrodes in the MoS<sub>2</sub>-based EDLTs have a nearly ohmic contact property.

With the secured thermally tolerable characteristics of the ILs, we measured the temperature-dependent current flows of the MoS<sub>2</sub> EDLTs in order to render the spectral DOS in the band gap. In fact, the temperature-dependent experiments of the device are typically performed at cryogenic temperatures. However, for large bandgap semiconductors such as MoS<sub>2</sub>, a Schottky barrier may be more prominent at low temperatures, resulting in increased contact resistance.<sup>17</sup> Thus, measurements of the temperature-dependent transport of the EDLTs above room temperature are more suitable in order to ensure precise extractions. Figure 3a shows the transfer curves with elevating temperatures from 300 to 400 K and steps of 25 K. When the temperature is raised, the intrinsic concentration of charge carriers in the channel increases, and the threshold voltage shifts to lower gate potentials than has been previously reported for conventional thin film transistors.<sup>7</sup> Even at the high temperature of 400 K, the MoS<sub>2</sub> EDLTs still were in the accumulation mode. The device characteristics, such as the transfer curve shape, the on-currents, and the on/off ratios, remained intact within the measurement range of the temperature.

Figure 3b, which shows the temperature dependent  $I_{DS}$  extracted from Figure 3a, shows the thermally activated drain currents ( $I_D$ ) of the MoS<sub>2</sub>-based EDLTs, where the currents can be described by

$$I_D = I_{D0} \exp(-E_a/kT) \quad (2)$$

where  $I_{D0}$  is the prefactor,  $E_a$  is the activation energy,  $k$  is the Boltzmann constant, and  $T$  is the temperature. In particular,  $I_{D0}$  and  $E_a$ , which are extracted from the  $\log(I_D)$  vs  $1/kT$  plots, can be varied with  $V_G$  as shown in Figure 3c and have a correlation known as the Meyer-Neldel (MN) rule:<sup>18,19</sup>

$$I_{D0} = I_{D00} \exp(E_a/E_{MN}) \quad (3)$$

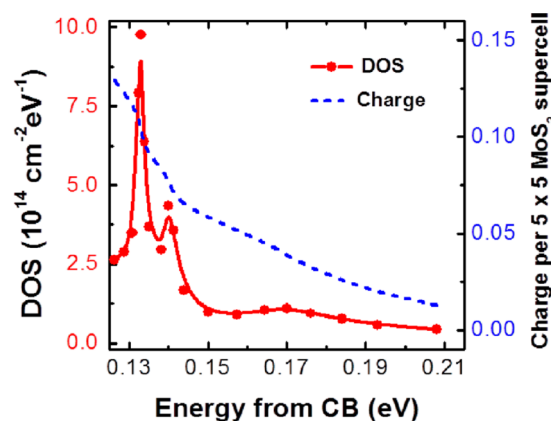
and this relation leads to the following equation:

$$I_D = I_{D00} \exp[(1/E_{MN} - 1/kT)E_a] \quad (4)$$

where  $I_{D00}$  is the characteristic constant and  $E_{MN}$  is defined as the MN energy, which is the common cross point shown in Figure 3b. Figure 3d shows that  $1/E_{MN}$  for our MoS<sub>2</sub>-based EDLTs is valid at  $9.03 \text{ eV}^{-1}$  in a range of activation energies from 0.14 to 0.21 eV, and  $I_{D00}$  is  $\sim 0.157$  mA. Collectively, these MN relations in the MoS<sub>2</sub> EDLT device indicate that the Fermi level,  $E_F$ , is varied by  $V_G$  despite the introduction of the defect states of the MoS<sub>2</sub> semiconductor surfaces. Consistent with this, the energy difference between  $E_F$  of the semiconductor and the band edge at the insulator-semiconductor interface is approximated with the measured  $E_a$  ( $V_G$ ) of  $I_D$  and can be expressed as  $E_a = E_F - E_C - qV_i$ , where  $V_i$  is the potential right at the IL contact with the MoS<sub>2</sub> semiconductor, which is close to zero.<sup>20,21</sup> As  $V_G$  increases, the induced carrier density per unit area corresponding to  $n_{2D} = (C_{EDL}\Delta V_G)/q$  increasingly fills the localized states from a lower energy level. Therefore, the increased number of charge carriers induced in the MoS<sub>2</sub> surface decreases  $E_a$ . In consideration that the MoS<sub>2</sub> layers were mechanically detached from bulk MoS<sub>2</sub> crystals, and thus the induced charge density should be much larger than the trap sites, we can estimate the induced surface charge DOS of the MoS<sub>2</sub>,  $n_{2D}(E)$ , to be

$$n_{2D}(E) = \frac{C_{EDL}}{q} \left( \frac{dE_a}{dV_G} \right)^{-1} \quad (5)$$

Figure 4 shows  $n_{2D}(E)$  and  $n_{2D}$  for a  $5 \times 5$  MoS<sub>2</sub> supercell as a function of the corresponding energy level.<sup>22</sup> An abrupt



**Figure 4.** Surface charge density of states of the MoS<sub>2</sub> in the EDLT devices and total charge concentration per  $5 \times 5$  MoS<sub>2</sub> supercell corresponding to the energy level from the conduction band.

decrease is observed in the  $n_{2D}$  values to  $\sim 0.14$  eV below the conduction bands, and the monotonous reduction of the values from the crossed level of the two distinct  $n_{2D}$  function lines. This relatively short band edge region can also be the reason for the near-ideal subthreshold swings of the MoS<sub>2</sub> EDLT devices. On the other hand, in contrast to the traces of  $n_{2D}$ , there are some peaks that individually represent their energy states with respect to the surface DOS ( $\text{cm}^{-2} \text{eV}^{-1}$ ). The most interesting peaks are measured at 0.135 and 0.145 eV below the conduction band, with the largest DOS values of  $9.75 \times 10^{14}$  and  $4.33 \times 10^{14} \text{ cm}^{-2} \text{eV}^{-1}$ , respectively. At the moment, the origin of these distinct peaks is unclear, but if we pay attention to the fact that the bandgap of the MoS<sub>2</sub> nanosheet strongly depends on the number of MoS<sub>2</sub> layers, a possible explanation



may be the fact that the monolayer MoS<sub>2</sub> nanosheets coexist with the MoS<sub>2</sub> multilayer in the channel area of the EDLT.<sup>23</sup>

#### 4. CONCLUSION

In conclusion, we have fabricated high-performance MoS<sub>2</sub>-based EDLT devices with high thermal stability, which allow for the performance of the variable temperature measurements on MoS<sub>2</sub>-based EDLT. Observed temperature-dependent behavior of activation energy for electron transport confirmed that the MN rule can be applied to the charge transport in the MoS<sub>2</sub> layer under high charge density. Furthermore, the 2D DOS profiles in the MoS<sub>2</sub> nanosheets were rigorously calculated by the stable measurements on the temperature-dependent I–V characteristics. The extracted surface DOS of the MoS<sub>2</sub> nanolayer shows two featured peaks at 0.135 and 0.145 eV below the conduction band, which suggest that the different energy levels are overlaid in the sub-bandgap, owing to the coexistence of monolayer and multilayers in the MoS<sub>2</sub> nanocrystals.

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##### Notes

The authors declare no competing financial interest.

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